

# Fabrication of a-CN<sub>x</sub>:H films with high nitrogen content by RF-plasma CVD of the CH<sub>3</sub>CN/N<sub>2</sub> gas mixture

CH<sub>3</sub>CN/N<sub>2</sub>混合気体の高周波プラズマCVDを用いた高窒素含有 a-CN<sub>x</sub>:H薄膜の形成

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Hydrogenated amorphous carbon nitride (a-CN<sub>x</sub>:H) films were formed by using a parallel-plate radio-frequency (RF) plasma chemical vapor deposition (CVD) of the gas mixture of CH<sub>3</sub>CN and N<sub>2</sub> where Ar was used as a buffer gas. The total pressure of Ar+N<sub>2</sub> were 0.3 Torr, whereas the partial pressure of CH<sub>3</sub>CN was suppressed to <10 mTorr. The compositional analysis was made by X-ray photoelectron spectroscopy (XPS), yielding [N]/([N]+[C])=0.30-0.45. The chemical structure of films were analyzed by using infrared absorption, Raman, and a high-resolution XPS.

## 1. Introduction

In the fabrication of thin films of carbon nitrides, it has been the central problem to incorporate N atoms into the C-C network of films as much as possible [1]. The reason for this chemical design is as follows. Since the length of the C-N bond (1.48 Å) is shorter than that of the sp<sup>3</sup>-hybridized C-C bond (1.54 Å) [2], high mechanical hardness is expected if the network of C-N bond is developed. In the well-known method of plasma-enhanced chemical vapor deposition (PECVD) using radio-frequency (RF) discharge of the gas mixture of N<sub>2</sub> and organic materials such as CH<sub>4</sub>, the nitrogen contents or the [N]/([N]+[C]) ratios of films have been suppressed to be <0.1 [3]. Therefore, the methodology to incorporate N atoms has been required. In this study, we have developed a method for this purpose using conventional parallel-plate RF-plasma CVD using the gas mixture of Ar+N<sub>2</sub>+CH<sub>3</sub>CN.

## 2. Experiments

Fig. 1 shows the schematic diagram of the apparatus of RF-plasma CVD. It consists of a stainless-steel chamber of the diameter of 254 mm. A pair of electrodes was placed inside of the chamber. Ar and N<sub>2</sub> were introduced with the total pressure of 0.3 Torr, where their partial pressures,  $P_{Ar}$  and  $P_{N_2}$ , respectively, were varied as listed in Table 1. A 13.56 MHz RF power of 50 W (Tokyo High Power, A-500) was applied to the upper electrode through a matching box (Tokyo High Power, HC-2000S). The lower electrode was grounded. Then, CH<sub>3</sub>CN was introduced from a Pyrex-glass nozzle whose orifice diameter was ≈1

mm. The partial pressure of CH<sub>3</sub>CN was <10 mTorr, although precise measurement was impossible because it was much lower than that of Ar or N<sub>2</sub>. Films were deposited onto Si substrates placed both onto upper and lower electrodes. The deposition time was 3 h.

Films were analyzed by use of Fourier-transformed infrared (FTIR) absorption spectroscopy (Nicolet 6700), Raman scattering (Renishaw 2000), and X-ray photoelectron spectroscopy (XPS) (JEOL JPS-9010).

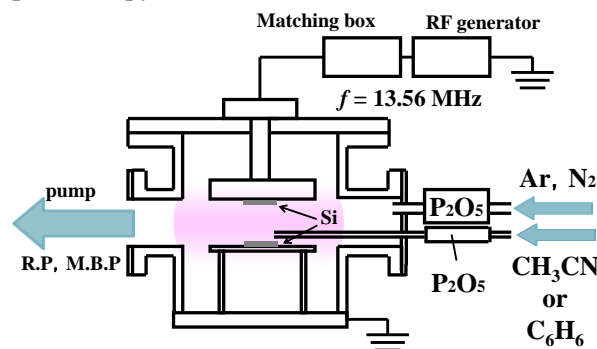


Fig. 1 RF-plasma CVD apparatus.

## 3. Results and discussion

Table 1 lists the results of the compositional analysis. In addition to the C and N atoms, significant amount of O atoms were detected. The [N]/([N]+[C]) ratios are plotted against  $P_{N_2}$  as shown in Fig. 2. These ratios exhibit 0.30-0.45, being much higher than the previous reports. Figs. 3(a) and 3(b) show the high-resolution XPS spectra of the C(1s) and N(1s) states, respectively, for the films formed on the lower electrode under the condition of  $P_{N_2}=0.1$  Torr. The C(1s) spectra were

Table 1 Experimental conditions and compositional analysis

$P_{Ar}$	$P_{N_2}$	C	N	O
[Torr]				
0.3	0.0	61.0±1.0	26.0±1.0	13.0±0.5
0.2	0.1	57.5±0.5	37.0±0.0	5.5±0.5
0.1	0.2	51.0±2.0	43.5±2.5	5.5±0.5
0.0	0.3	59.5±0.5	33.5±2.5	7.0±3.5

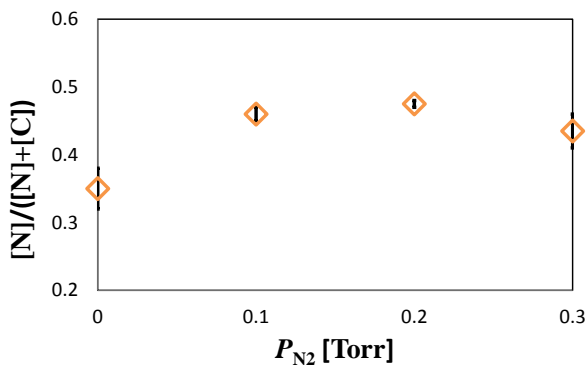


Fig. 2 [N]/([N]+[C]) ratios of films.

fitted satisfactorily by Gaussian profile of C=N bond with smaller contributions of C=C and C-C bonds [4]. The N(1s) peak was fully explained by a single Gaussian profile of N=C bond [5]. These results indicate that the network of films were

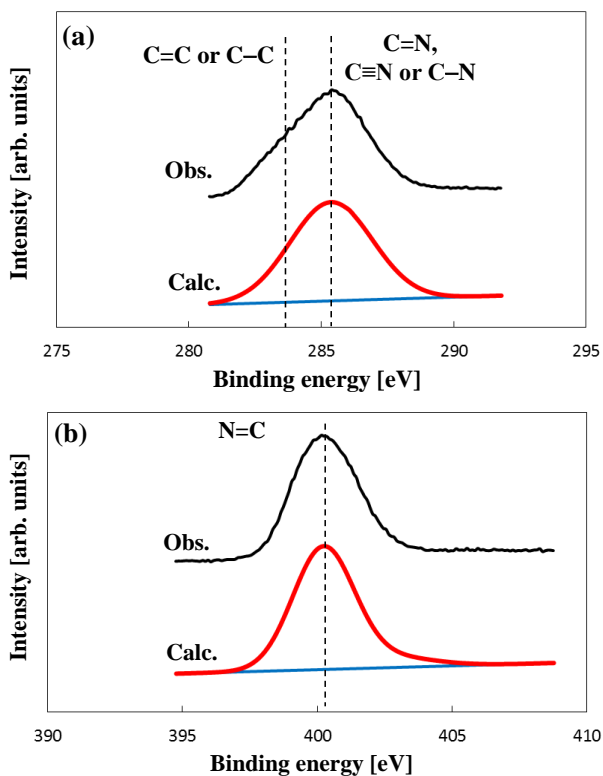


Fig. 3 High resolution XPS spectra (upper trace) and the result of the simulation analysis (lower trace). (a) C(1s). (b) N(1s).

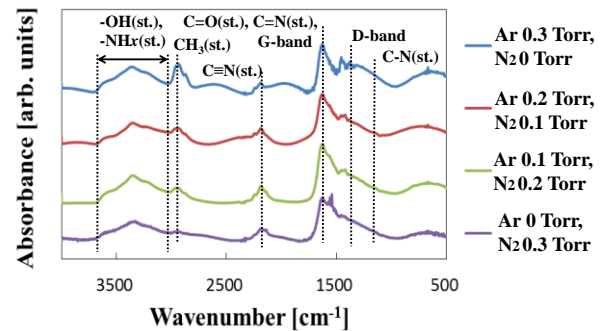


Fig. 4 FTIR spectra.

dominated by C=N bonds; the incorporation of  $N_2$  gas was negligible. Fig. 4 shows the FTIR spectra of the films deposited onto the upper electrode. The G- and D-bands were observed in the region of  $\approx 1550$  and  $\approx 1400$   $cm^{-1}$ , respectively [6], where the G-band may be overlapped by C=N stretching vibration. With the consideration of the result of XPS, the assignment of C=N seems to be correct for the  $\approx 1550$   $cm^{-1}$  band.  $NH_y$  ( $y=1,2$ ) stretching vibration was observed around  $3300$   $cm^{-1}$ . The  $C\equiv N$  stretching vibration at  $2300$   $cm^{-1}$  was rather weak. In addition, weak C-N stretching vibration was observed around  $1300$   $cm^{-1}$ . All these vibration modes were almost independent of  $P_{N_2}$ , whereas the  $CH_z$  ( $z=1-3$ ) stretching vibration modes observed around  $2800$   $cm^{-1}$  were dependent on.

#### 4. Conclusion

This study reports on the novel methodology to fabricate  $a-CN_x:H$  films with high nitrogen content. The essential point may be the suppression of the partial pressure of organic vapor compared to the total pressure. The chemical structure of films may dominate the C=N network. The  $CH_z$  and  $C\equiv N$  groups usually observed in the films with low nitrogen content decrease their densities in the present films.

#### References

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